

Parameters for the growth of single-phase In-rich $\text{In}_x\text{Ga}_{1-x}\text{N}$ ($x > 0.5$) in MOVPE

Y. Nakagawa, K. Koide, A. Hashimoto, and A. Yamamoto

Dept. of Electrical and Electronics Engineering, Faculty of Engineering, Fukui University

3-9-1 Bunkyo, Fukui 910-8507, Japan

E-mail: e981561@icpc00.icpc.fukui-u.ac.jp

The group nitride InGaN alloys, especially Ga-rich InGaN, have recently attracted much attention as potential materials for the fabrication of green~violet light-emitting diodes and injection lasers. In-rich InGaN, on the other hand, have been expected as promising materials for high speed and high power electronic devices because of the high electron mobility in InN. Compared with Ga-rich InGaN, however, growth of In-rich InGaN has not yet been widely studied. For MOVPE of InGaN, there have been reported several parameters which governs In mole fraction incorporated in solid. Growth temperature is one the most dominant factors, as described by Matsuoka et al. [1]: a growth temperature higher than 700 °C markedly reduces In mole fraction. Koukitu et al. [2] theoretically predict that In solid composition in InGaN is decreased with increasing H_2 partial pressure in the growth atmosphere. This prediction is supported by the experimental result that InGaN with whole In composition is obtained at 680 °C by using plasma-excited MOVPE [3], where NH_3 which makes H_2 is not used. In this paper, we report MOVPE growth of In-rich InGaN grown at 700 °C, and discuss effects of growth temperature and V/III ratio on the growth of In-rich InGaN

InGaN films were grown on $\text{-Al}_2\text{O}_3(0001)$ substrate by MOVPE with a horizontal reactor. The growth was conducted at a reduced pressure (70 Torr) using TMIn, TEGa and NH_3 as In, Ga and N sources, respectively. N_2 was used as a carrier gas. Substrates were cleaned in H_2 at 950 °C and then nitrided at 900 °C in an ambient of NH_3 before the growth. InGaN films were grown at 550 and 700 °C with a thickness of 0.2~0.4 μm . The θ -2 θ X-ray diffraction pattern was measured to get information about In content and phase separation in grown InGaN films. The grown films were also characterized with RHEED, AFM and Hall measurements.

Figure 1 shows θ -2 θ X-ray diffraction patterns for $\text{In}_x\text{Ga}_{1-x}\text{N}$ films grown at 700 °C. As seen in this figure, multi-peak diffraction pattern is obtained when $\text{TMIn} / (\text{TMIn} + \text{TEGa})$ exceeds 0.75, showing that phase separation occurs. This is in contrast to the case for films grown at 550 °C [4], where single-peak diffraction pattern is obtained for $0.5 \leq x \leq 1$. Figure 2 shows the map of X-ray diffraction pattern, single- or multi-peak for InGaN films grown at 700 °C in the plane of $\text{TMIn} / (\text{TMIn} + \text{TEGa})$ vs. V/III ratio. One can see that boundary between single- and multi-peak diffractions is dependent not only on $\text{TMIn} / (\text{TMIn} + \text{TEGa})$ but also on V/III ratio. The region of single-peak diffraction seems to be widened with increasing V/III ratio. It is also pointed out that In solid composition in InGaN is decreased with increasing V/III ratio. This is attributed to an increased partial pressure of H_2 , which is produced by NH_3 decomposition, due to the increased V/III ratio, as predicted [2]. Figure 3 shows $\text{TMIn} / (\text{TMIn} + \text{TEGa})$ dependence of In solid composition. Some typical data ever reported for InGaN are also included in the figure. The results obtained in this work are similar to those by plasma-excited MOVPE at 680 °C [3]. Figure 4 shows RHEED patterns for $\text{In}_x\text{Ga}_{1-x}\text{N}$ with $x = 0.5 \sim 0.6$ grown at 550 or 700 °C. The film grown at 550 °C shows a characteristic RHEED pattern showing that the film contains a mixture of wurtzite and twinned zincblende crystals [4]. Such a pattern is not observed for the film grown at 700 °C.

In summary, MOVPE growth of InGaN was performed and parameters for the growth of single-phase In-rich InGaN, especially $\text{TMIn} / (\text{TMIn} + \text{TEGa})$ and V/III ratio, were discussed. Boundary between single-phase and multi-phase growth was found to be dependent not only on $\text{TMIn} / (\text{TMIn} + \text{TEGa})$ but also on V/III ratio. In solid composition in InGaN was decreased with increasing V/III ratio. This seems to be due to the increased H_2 partial pressure in the growth atmosphere. The InGaN film grown at 700 °C was found not to

contain the mixture of wurtzite and twinned zincblende crystals observed for the film grown at 550 °C.

REFERENCES

- [1] T. Matsuoka et al. J. Electron. Mater. 21, 157 (1992).
- [2] A. Koukitu et al. J. Cryst. Growth 197, 99 (1999).
- [3] T. Tokuda et al. J. Cryst. Growth 187, 179 (1998).
- [4] A. Yamamoto et al. phys. stat. sol. (a) 176,237 (1999)

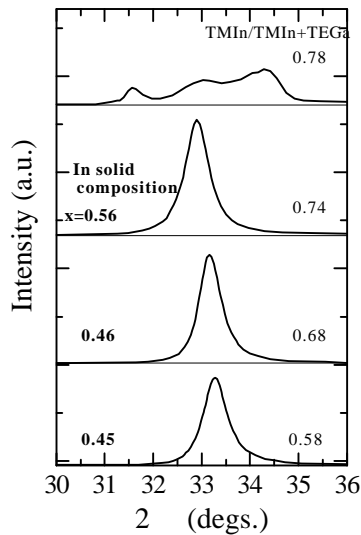


Fig.1. X-ray diffraction patterns for InGaN films grown at 700 °C.

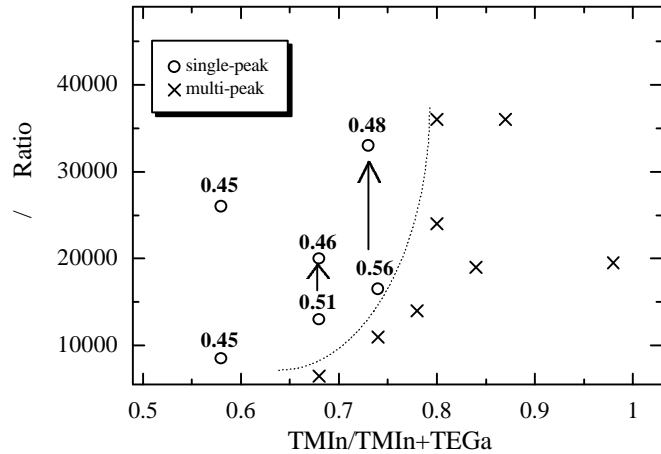


Fig.2. The map of X-ray diffraction pattern, single- or multi-peak for InGaN films grown at 700 °C in the plane of TMIn / (TMIn + TEGa) vs. I / Ratio. Numbers in the figure are In solid compositions in InGaN.

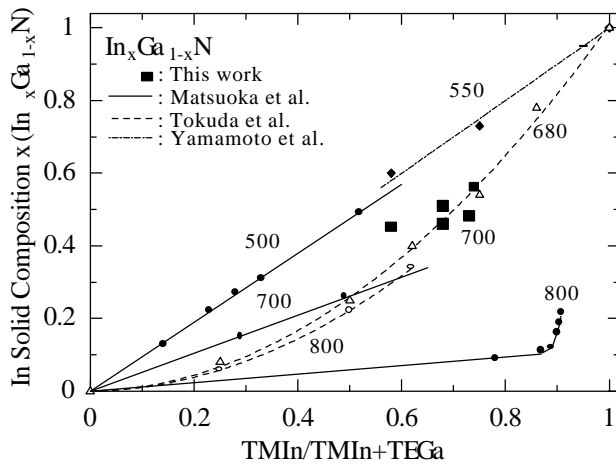


Fig.3. TMIn / (TMIn + TEGa) dependence of In solid composition in InGaN. Typical data ever reported for InGaN are also shown in the figure.



(a) 550



(b) 700

Fig.4. RHEED patterns for $\text{In}_x\text{Ga}_{1-x}\text{N}$ with $x = 0.5 \sim 0.6$ grown at 550 and 700 °C.